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# Colloids and Surfaces A: Physicochemical and Engineering Aspects

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## Zeolite A grown on fiberglass: A prominent CO<sub>2</sub> adsorbent for CO<sub>2</sub>/CH<sub>4</sub> separation

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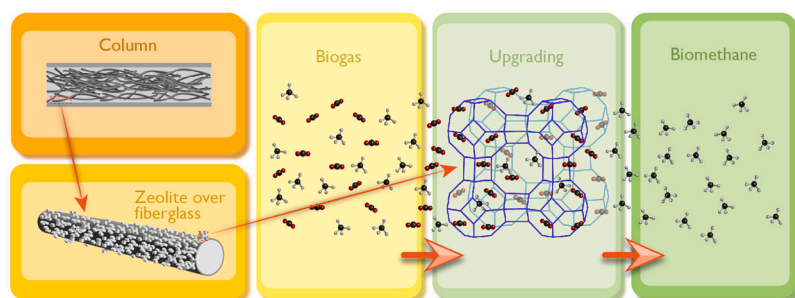
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### GRAPHICAL ABSTRACT



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### ABSTRACT

Considerable efforts have been devoted to developing adsorbents capable of retain CO<sub>2</sub> either via direct capture from air, or via more sophisticated systems such as biogas, a mixture from which methane can be upgraded. In this work, we present an experimental investigation of zeolite NaA, grown over alkaline-treated fiberglass, synthesized by means of hydrothermal route and tested as adsorbent for CO<sub>2</sub> removal in the CO<sub>2</sub>/CH<sub>4</sub> mixture. Zeolite NaA was successfully obtained, with absence of secondary phases, as detailed by the characterization results. Adsorption experiments, presented in terms of isotherms, demonstrated the potential of the synthesized material as a selective CO<sub>2</sub> adsorbent with potential to be applied for biogas enrichment.

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## 1. Introduction

In the face of the growing demand for renewable energy sources and the environment impacts related to the process of global warming, the development of new technologies for CO<sub>2</sub> capture has become a critical environmental issue, and certainly one of the greatest challenges of the humankind [1]. In this regard, the degradation of urban solid wastes leads to the production of huge amounts biogas, a gaseous mixture composed mainly of methane (CH<sub>4</sub>) and carbon dioxide (CO<sub>2</sub>), and other minor components such as H<sub>2</sub>S, siloxanes, N<sub>2</sub> and water vapor [2]. CH<sub>4</sub> is known to promote harmful impacts on the environment, considering its contribution for the global warming, at least 20 times more aggressively than CO<sub>2</sub> [3]. On the other hand, the CH<sub>4</sub> present in the biogas is a versatile energy source and therefore has a potential to be harnessed and applied in many different fields replacing fossil fuel in electric power generation, vehicle engines, heat production etc.

Nevertheless, the use of biogas as a fuel requires previous treatments aiming the removal of H<sub>2</sub>O and H<sub>2</sub>S, and the capture of CO<sub>2</sub>, a process commonly referred as biogas upgrading [4]. H<sub>2</sub>O and H<sub>2</sub>S can be removed via relatively simple and low-cost methods, whereas CO<sub>2</sub>, in particular for being present in considerable higher concentrations (30–60%), usually demands more robust technologies for its removal. The leading techniques employed for CO<sub>2</sub> removal are physical absorption (washing) [5–7], chemical absorption [8], membrane separation [9] and pressure swing adsorption (PSA) [10]. Other incipient techniques include cryogenic separation [11], in situ enrichment [12], hydrates formation [13], biological methods [14], and carbon mineralization [15]. Among these techniques, PSA stands out given the relatively low costs of operation and the non-generation of residual solvents. The PSA efficiency depends largely on the adsorbent choice, which should be selective towards CO<sub>2</sub>, present high adsorption capacity and the ability to undergo regeneration [16,17]. Besides, the adsorbent should be thermally stable, chemically inert, and present high surface area. In this regard, materials such as activated carbons and microporous aluminosilicates [18,19], with emphasis on zeolites [20], have been widely employed in biogas upgrading.

Zeolites are defined as silica-based microporous materials, in which some Si atoms are replaced by other elements as Al (mainly), P, Fe, B, Ga, Ti, Ge etc. Zeolites are characterized by a 3D structure resulting from the bonds between the tetrahedral TO<sub>4</sub> units (T = Si, Al, P, Fe etc.), linked via oxygen atoms. The zeolitic structures present channel and cavity systems with molecular dimensions, capable of reversibly induce the addition and removal of guest compounds, what make them excellent adsorbents [21–23]. Despite all the appealing properties, zeolites possess one important drawback which is associated with their polycrystalline nature, i.e., their powder aspect. For practical applications, zeolites are usually required to be packed as pellets, dramatically reducing the internal available area and therefore their performance. Alternatively, zeolites can be prepared, either associated or not with other materials, in hierarchical structures, in such a way that their crystals are arranged to form interparticle mesopores, or even macropores. Several strategies can be adopted in the preparation of hierarchical zeolites: (i) mesoporosity formed from the controlled degradation of zeolite crystals [24,25], (ii) interparticle mesopores by the arrangement of nanozeolite crystals [26], (iii) zeolite crystallization by using bifunctional surfactants [27–29] and (iv) the crystallization of zeolites over other materials (zeolitization) [30,31], including membranes, a growing field in separation processes [32,33]. Zeolitization can be accomplished by growth, dispersion, or aggregation of zeolitic crystals over either inorganic or organic materials, leading to formation of bigger pores, to which the zeolite micropores will be connected, as a result of the particles organization. Several different materials have been used as support for zeolite growth, particularly with monolithic features [34–36]. However, materials such as fiber glass can offer certain advantages from mechanical point of view given its malleability.

Herein, we present the preparation of a zeolite composite, with

hierarchical features, by coating activated fiberglass with zeolite crystals, grown via hydrothermal route, and the assessment of its performance as CO<sub>2</sub> adsorbent in CO<sub>2</sub>/CH<sub>4</sub> mixtures.

## 2. Experimental

### 2.1. Zeolite A synthesis

Zeolite A-fiberglass was prepared by using fiberglass, FG, (Synth, 99.5% SiO<sub>2</sub>) which underwent an alkaline treatment targeting generation of roughness so that further zeolite nucleation could take place, as reported by Silva et al. [37]. This treatment consisted of immersing 10.0 g of FG in 100.0 mL of NaOH 4.0 mol L<sup>-1</sup> in a polypropylene flask, and mechanically stirred for 72 h (200 rpm). The activated FG sample (aFG) was then washed with HCl 1.0 mol L<sup>-1</sup> (2 ×), and with distilled water (2 ×), and finally dried at 80.0 °C for 24 h.

A precursor solution for zeolite synthesis was prepared by dissolving 7.16 g of Na<sub>2</sub>SiO<sub>3</sub> (Sigma-Aldrich, 50–53% SiO<sub>2</sub>) in 35.0 mL of NaOH 0.21 mol L<sup>-1</sup>, in a polypropylene beaker, and 5.00 g of NaAlO<sub>2</sub> (Sigma-Aldrich, 50–56% of Al<sub>2</sub>O<sub>3</sub>) with 35.0 mL of NaOH 0.21 mol L<sup>-1</sup> in another beaker, and then these two solutions were mixed. 8.00 g of the sample aFG was transferred to a Teflon-lined stainless-steel autoclave, together with the precursor solution, reaching the complete volume of the autoclave (72 mL). The system was kept at 25.0 °C for 18.0 h and then heated at 100 °C for 4 h under autogenous pressure and static conditions. After cooling, the obtained material was washed with distilled water and centrifuged at 4000 rpm (7 ×). To eliminate zeolite crystals which were not formed incorporated onto the fiberglass, water splash was applied to the samples, spread over a sieve as screen protection. This way, it is ensured that the final material is robust to withstand flow either of liquids or gases without having its structure compromised. The sample was named as aFG-Zeo. For comparison purposes, conventional zeolite NaA (Zeolite A) was prepared following the method described by Thompson et al. [38].

### 2.2. Characterization

X-ray powder diffraction (XRPD) patterns of the samples FG, aFG, aFG-Zeo, and Zeolite A, were obtained by using a Panalytical (X-Pert Pro MPD) x-ray powder diffractometer. The powder patterns were collected in the continuous mode ( $\theta$ – $\theta$  scan) with step size of 0.013°(2 $\theta$ ). Monochromatic Co-K $\alpha$ 1 radiation was used, obtained with the tube operating at 40 kV and 40 mA.

Fourier transform infrared (FTIR) spectra of samples in the form of KBr pellets were obtained, at room temperature, on a Shimadzu IRealise FTIR spectrometer in the 4000–400 cm<sup>-1</sup> region, with a nominal resolution of 2 cm<sup>-1</sup>.

For scanning electron microscopy (SEM), a Quanta 200 (FEI) microscope was used, by employing detector for secondary electrons and energy-dispersed X-ray. Samples were dispersed over a conductive carbon double-sided sticky tape, on aluminum supports. To reduce charging effects, the samples were coated with a thin layer of gold (ca. 20 nm).

CO<sub>2</sub> adsorption isotherms were obtained at 273 K in an Autosorb-iQ<sub>3</sub> equipment (Quantachrome, USA) and used to characterize the micropore volume of the studied zeolites. The samples (ca. 0.1 g) were previously degassed under the following conditions: heating under high vacuum (10<sup>-4</sup> mbar), from room temperature to 180 °C, at a heating rate of 2 °C min<sup>-1</sup>, maintaining at the final temperature for a period of 6 h. The microporous volume was calculated using the Dubinin-Radushkevich equation.

### 2.3. Adsorption experiments

Isotherms of pure CH<sub>4</sub>, pure CO<sub>2</sub>, and a 47% CH<sub>4</sub>/53% CO<sub>2</sub> (v/v) were obtained at 25 °C up to 10 bar using a magnetic suspension balance

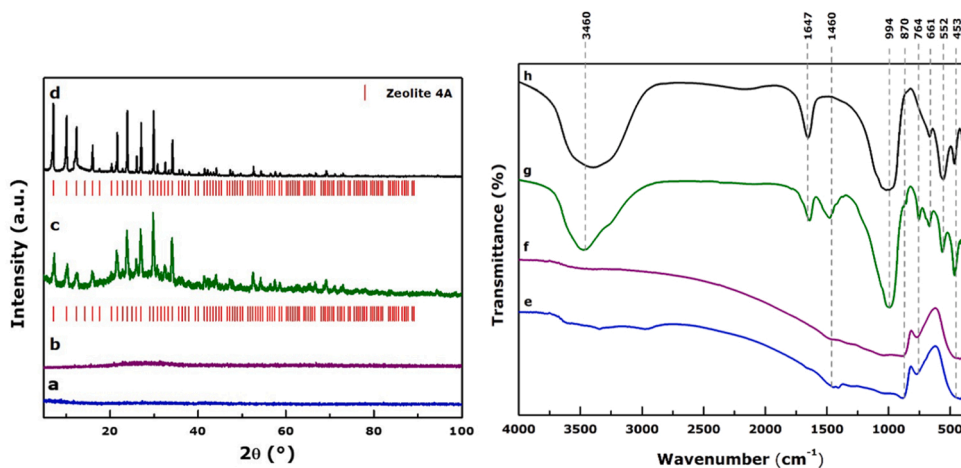


Fig. 1. Co-K $\alpha$  XRD patterns for the samples (a) FG, (b) aFG, (c) aFG-Zeo, and (d) powder zeolite A. FTIR, in KBr, for the samples (e) FG, (f) aFG, (g) aFG-Zeo, and (h) powder zeolite A.

Rubotherm (Bochum, Germany), equipped with a mixture unit. Before the equilibrium essays, the samples (ca. 0.5 g) were thermally treated at 300 °C with heating rate of 2 °C min<sup>-1</sup>, under vacuum, for 10 h.

The adsorption isotherms data were fitted to Langmuir and Sips isotherm models, Eqs. 1 and 2.

$$q_{Lang} = \frac{q_{max}bP}{1 + bP} \tag{1}$$

$$q_{Sips} = \frac{q_{max}(bP)^n}{1 + (bP)^n} \tag{2}$$

Where  $q$  is the amount adsorbed at a partial pressure  $P$ ,  $q_{max}$  is the maximum amount that could be adsorbed,  $b$  is the parameter of affinity between adsorbate and adsorbent and  $n$  is a parameter of Sips model that characterize the heterogeneity of the system.

The CO<sub>2</sub>/CH<sub>4</sub> selectivity was calculated using the Eq. 3, which  $q_{CO_2}$  and  $q_{CH_4}$  were obtained from Langmuir and Sips models.

$$\alpha_{CO_2/CH_4} = \frac{q_{CO_2}/y_{CO_2}}{q_{CH_4}/y_{CH_4}} \tag{3}$$

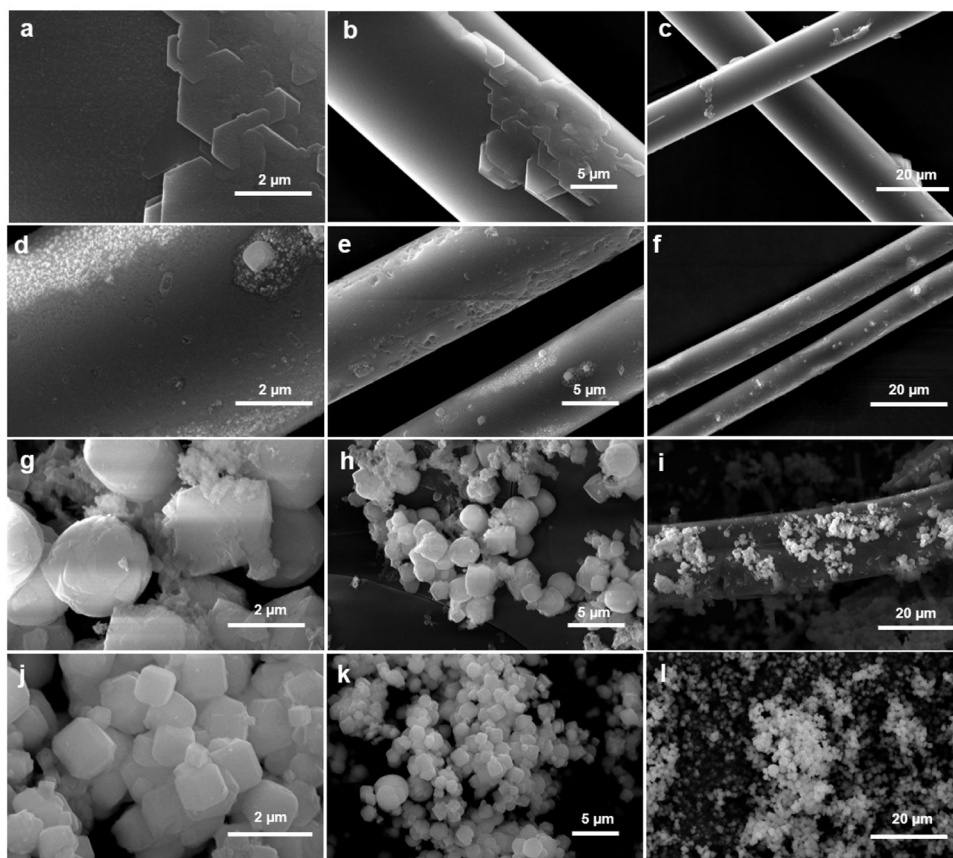


Fig. 2. SEM images of samples (a-c) FG, (d-f) aFG, (g-i) aFG-Zeo, and (j-l) zeolite NaA.

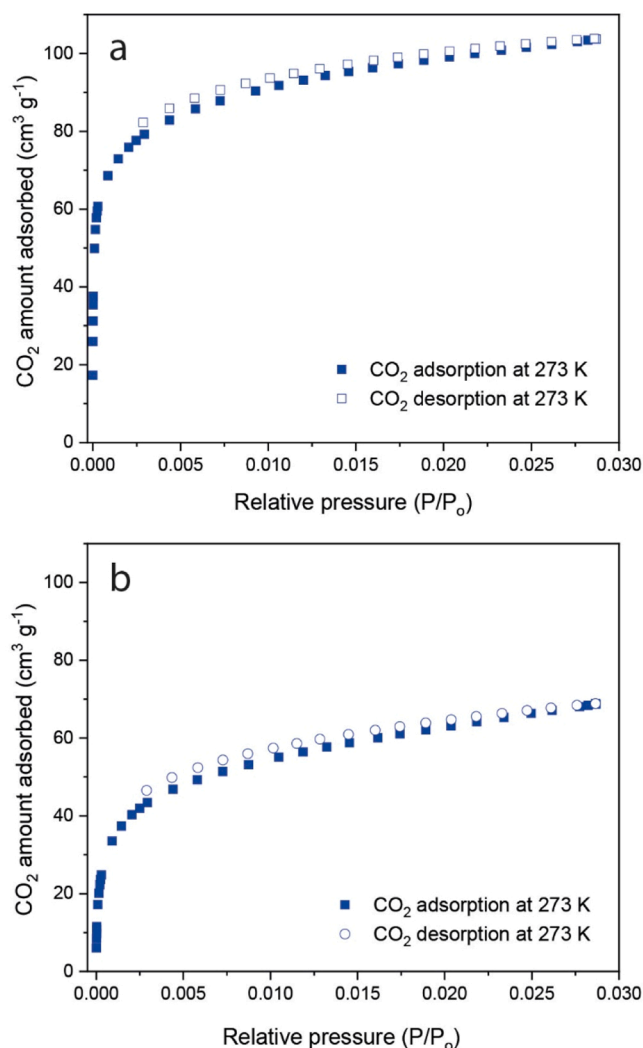


Fig. 3. Standard experimental isotherms of CO<sub>2</sub> (273 K) of (a) zeolite A and (b) glass fiber-zeolite A.

### 3. Results and discussion

#### 3.1. Characterization

XRPD patterns of the prepared samples are shown in Fig. 1, together with the FTIR spectra. The fiberglass, sample FG (Fig. 1a), underwent alkaline treatment aiming a surface partial deformation so that some homogeneous roughness could be generated, facilitating the nucleation of zeolite crystals in the subsequent synthesis stage. The XRD pattern of the alkaline activated sample, aFG, (Fig. 1b) indicates that no structural change occurred, with its amorphous nature been preserved. The intense diffraction peaks observed in Fig. 1c are consistent with zeolite A (ICSD 24901), indicating its crystallization, without other signals which could be associated with secondary phases. When the relative intensity of the peaks referent to the samples aFG-Zeo and Zeolite A are compared, it is observed that the first three peaks of the samples aFG-Zeo, with  $2\theta$  equal to 7.203°, 10.193°, and 12.492° (Fig. 1c) are less intense. Such characteristic is associated with a specific amount of zeolite A crystals generated over the fiberglass surface. The reduction in intensity of the peaks of a crystalline material on a glass surface has been observed in other studies involving fiberglass and is consistent with the zeolite synthesis over such material [39]. The observed background is result of the core fiberglass over which the zeolite crystals were formed.

FTIR provides invaluable information related to the structure of the

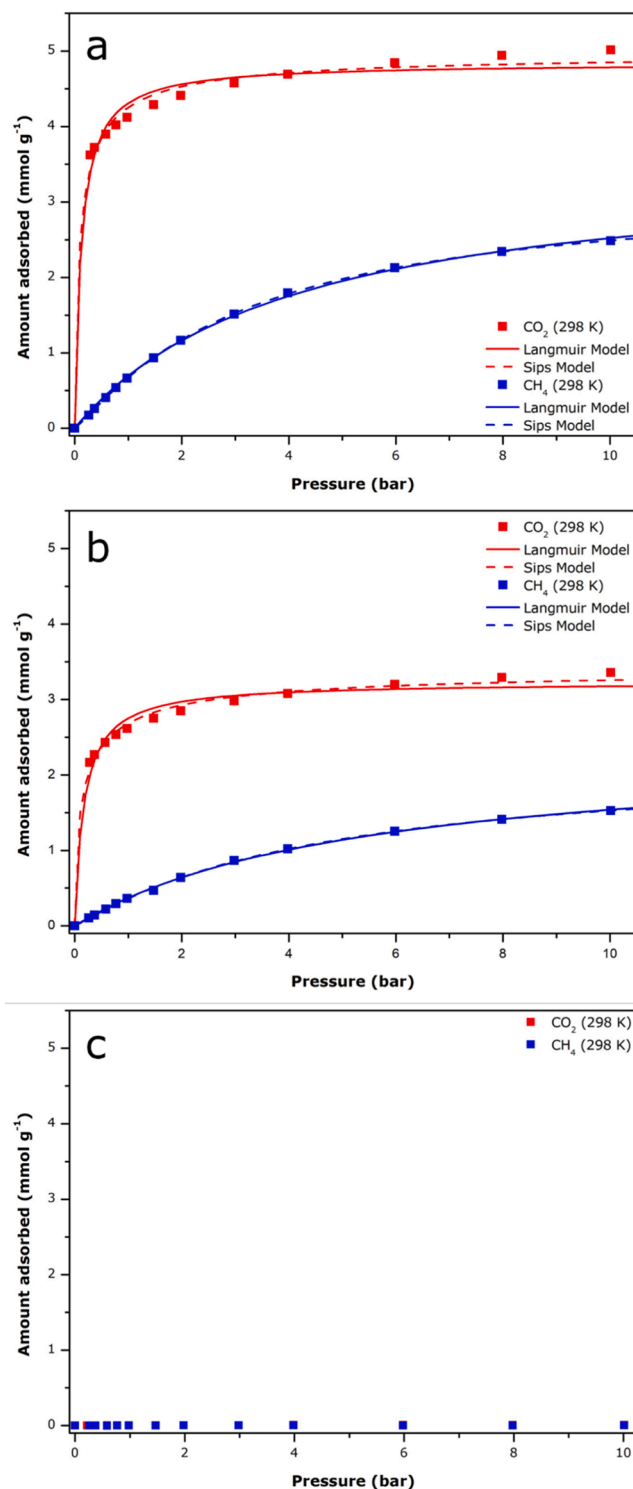


Fig. 4. Adsorption isotherms of CO<sub>2</sub> and CH<sub>4</sub> at 298 K for the samples (a) zeolite A, (b) aFG-Zeo, and (c) aFG. Langmuir and Sips equilibrium models were applied to the experimental data.

synthesized zeolite and can therefore be useful as a complementary tool for XRD data interpretation. The FTIR spectrum of zeolite A, presented in Fig. 1h, shows characteristic bands of the zeolite. The intense signal centered at 994 cm<sup>-1</sup> is consistent with Si(Al)-O asymmetric stretching [40]. The three signals below 990 cm<sup>-1</sup> are part of what is known as the fingerprint region. The band at 661 cm<sup>-1</sup> is associated with internal vibration of the (Si, Al)-O symmetric stretching. The band observed at 552 cm<sup>-1</sup> is related to the external vibration of the double four-rings

**Table 1**

Fitting parameters of Langmuir model and Sips model.

Langmuir	$q_m$ (mmol g <sup>-1</sup> )		$b$ (bar <sup>-1</sup> )		n	$R^2$	
	CO <sub>2</sub>	CH <sub>4</sub>	CO <sub>2</sub>	CH <sub>4</sub>		CO <sub>2</sub>	CH <sub>4</sub>
Zeolite A	4.84	3.55	8.09	0.24	-	0.99307	0.99963
aFG-Zeo	3.23	2.37	5.76	0.19	-	0.99126	0.99961
Sips	$q_m$ (mmol g <sup>-1</sup> )		$b$ (bar <sup>-1</sup> )		n	$R^2$	
	CO <sub>2</sub>	CH <sub>4</sub>	CO <sub>2</sub>	CH <sub>4</sub>		CO <sub>2</sub>	CH <sub>4</sub>
Zeolite A	4.99	3.19	9.73	0.31	0.77	1.12	0.99629
aFG-Zeo	3.46	2.17	6.44	0.22	0.67	1.07	0.99732

structure, which connects the  $\beta$ -cage, and the band at 453 cm<sup>-1</sup> corresponds to internal vibration of (Si, Al)-O bending [41]. The presence of structural water is confirmed by the large band at 3460 cm<sup>-1</sup> [41,42]. All these bands are also observed in the spectrum of the sample aFG-Zeo, the zeolite grown over the fiberglass (Fig. 1i), accompanied by the bands at 764 cm<sup>-1</sup>, and 870 cm<sup>-1</sup> and 1460 cm<sup>-1</sup>. These later are also present in the spectrum of the sample aFG and are related to the fiber structure.

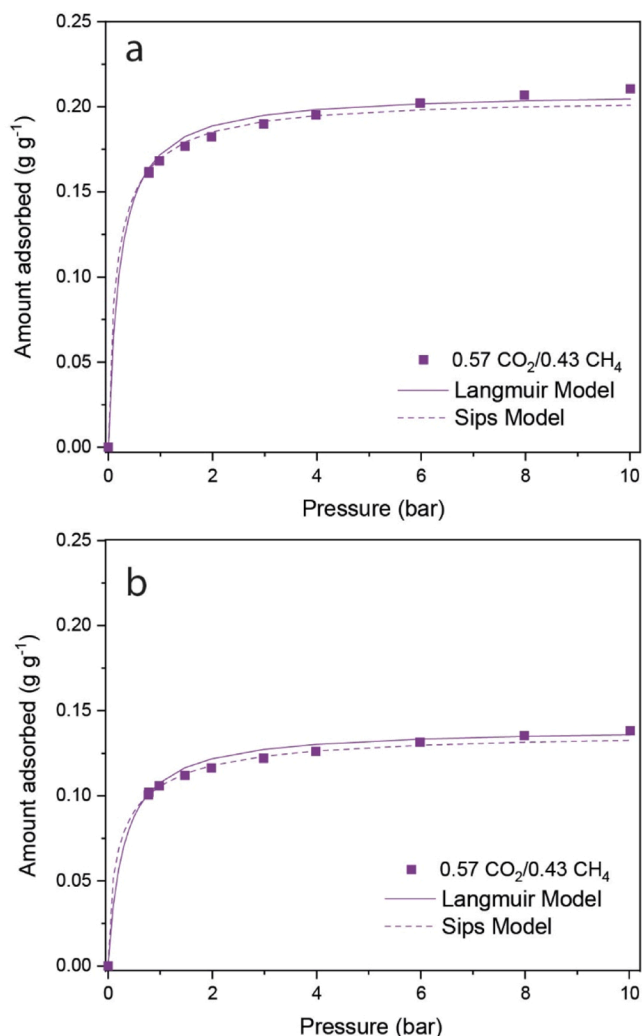
SEM images of all the samples are presented in Fig. 2. The sample FG (Fig. 2a-c) is composed of fibers with surface predominantly smooth, with uniform texture, which gives rise to a rough surface, but still in a controlled way, when submitted to alkaline treatment (Fig. 2d-f). The formation of zeolite crystals over the surface of the alkaline activated fiberglass sample is easily observable in Fig. 2g-i. The zeolite A typical crystals with cubic habit are formed directly over the activated fiberglass, with prevalence in regions of the fiberglass surface where the roughness is more evident. To emphasize the zeolite synthesis efficiency in presence of the fiberglass, zeolite A crystals, in the powder form, are shown in Fig. 2j-l, allowing direct comparison. The loading amount of zeolite over the fiberglass surface was estimated in 45–50%, based on the X-ray fluorescence analysis combined with the evaluation of a series of SEM images of the treated fiberglass, the pure zeolite, and fiberglass covered with the zeolite.

Textural properties such as surface area, pore volume and pore size distribution are essential to understand the porous structures of zeolitic materials. N<sub>2</sub> (at 77 K) and Ar (at 87 K) are the two most common gases used as probe. Nevertheless, their rate of diffusion at low temperature into zeolite A micropores limits their application in this sense [43]. Alternatively, CO<sub>2</sub> has been applied as a useful strategy for such cases, with isotherms obtained at 273 K, even though some characteristics related to the strong CO<sub>2</sub>-CO<sub>2</sub> interaction and the incipient theoretical background still possess some challenges for its effective application [42,44]. CO<sub>2</sub> adsorption isotherms for zeolite A and aFG-Zeo samples are presented in Fig. 3.

The specific micropore volume of zeolite A and aFG-Zeo samples calculated from CO<sub>2</sub> adsorption isotherms at 273 K are 0.23 cm<sup>3</sup>g<sup>-1</sup> and 0.17 cm<sup>3</sup>g<sup>-1</sup>, respectively. This reduction on the specific micropore volume occurs because the fiberglass does not contribute to the microporosity of the samples.

### 3.2. Adsorption isotherms

The isotherms of pure CH<sub>4</sub> and CO<sub>2</sub>, obtained for the samples of zeolite A, aFG-Zeo, and aFG are presented in Fig. 4. As expected, zeolite A presented high affinity for CO<sub>2</sub> at low pressures, as indicated by the rectangular shape of CO<sub>2</sub> isotherm (Fig. 4a), with adsorption capacity comparable to those verified in previous works [45,46]. The higher affinity towards CO<sub>2</sub> compared to CH<sub>4</sub> results from its higher critical temperature, i.e., CO<sub>2</sub> presents a condensable vapor behavior in the given conditions and is therefore less volatile than CH<sub>4</sub>, and so easier adsorbed [47]. The CO<sub>2</sub> adsorption capacity, at 10 bar and 298 K, decreases from ca. 5 mmol g<sup>-1</sup>, in sample zeolite A, to ca. 3 mmol g<sup>-1</sup> in sample aFG-Zeo (Fig. 4b), which can be seen as a satisfactory result considering that the later contains less zeolite. The fiberglass, over which the zeolite crystals are dispersed, does not contribute to CO<sub>2</sub>



**Fig. 5.** Adsorption biogas upgrading experiment with mixture 57% CO<sub>2</sub>/43% CH<sub>4</sub> for samples (a) zeolite A and (b) aFG-Zeo.

adsorption, as shown in Fig. 4c.

The fitting parameters of Langmuir model and Sips model for CO<sub>2</sub> and CH<sub>4</sub> adsorption isotherms on zeolite A and aFG-Zeo are presented in Table 1. Both models fit satisfactorily the experimental data, showing  $R^2 > 0.99$ , although the Sips model showed better fitting. The presence of fiberglass reduces the value of all parameters. The decrease in  $q_{max}$  value occurs because the fiberglass does not contribute to CO<sub>2</sub> and CH<sub>4</sub> on the studied conditions.

The isotherms of the mixture of 43% CH<sub>4</sub>/57% CO<sub>2</sub> (v/v) are shown in Fig. 5, together with the prediction of the extended models of Langmuir and Sips (Eqs. 4 and 5, respectively). As expected, both models fitting satisfactorily the experimental data for mixtures of CH<sub>4</sub> and CO<sub>2</sub>, showing  $R^2 > 0.99$ .

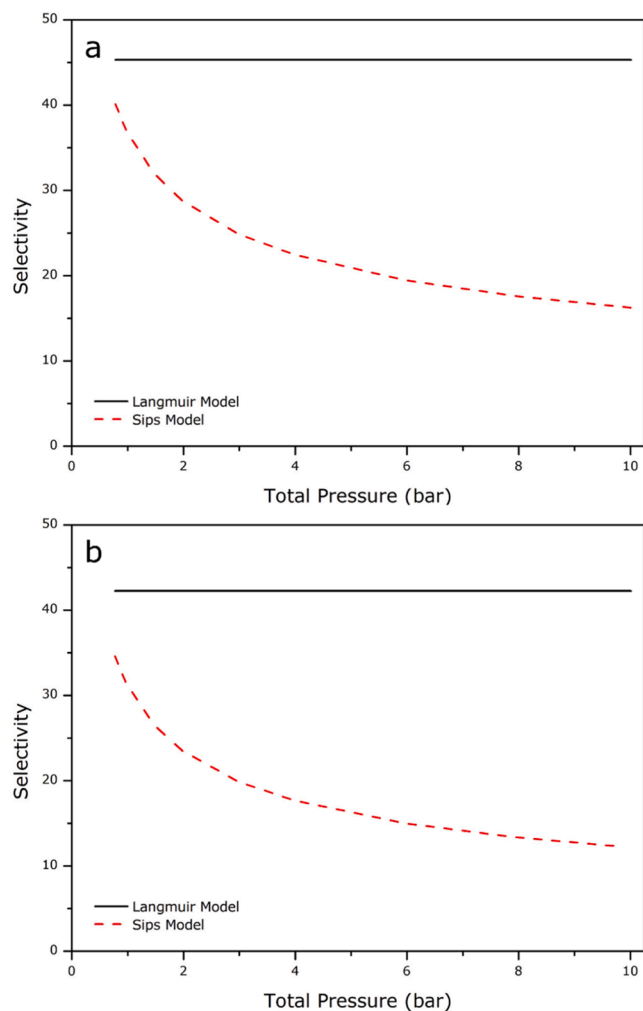


Fig. 6. CO<sub>2</sub>/CH<sub>4</sub> selectivity calculated using Langmuir model and Sips model for samples (a) zeolite A and (b) aFG-Zeo.

$$q_{i,Lang} = \frac{q_{\max,i} b_i P_i}{1 + \sum_{j=1}^N b_j P_j} \quad (4)$$

$$q_{i,Sips} = \frac{q_{\max,i} (b_i P_i)^n}{1 + \sum_{j=1}^N (b_j P_j)^n} \quad (5)$$

The selectivity of CO<sub>2</sub>/CH<sub>4</sub> (Fig. 6) was calculated using the predict amount adsorbed of each gas by Lagmuir model and Sips model. The Langmuir model gives a constant selectivity correspondent to the ratio between the affinity parameter of CO<sub>2</sub> and CH<sub>4</sub> for each sample, while the Sips model give a decreasing selectivity that varies between 40 and 16 for zeolite A, and 35 and 12 for aFG-Zeo sample. These results show that the fiberglass over which the zeolite crystals are grown reduce the selectivity less than 35%, which can be seen as acceptable considering its structural role, i.e., to support the zeolite crystals.

#### 4. Conclusion

Zeolite NaA was successfully synthesized over alkaline-activated fiberglass surface by employing a simple and low-cost strategy, with absence of secondary crystalline phases. The hydrothermal treatment applied to the fiberglass allowed a surface modification that resulted in controlled roughness, a condition that facilitated the growth of zeolite crystals. Adsorption essays showed that the obtained material presented

appreciable CO<sub>2</sub> adsorption capacity as well as high selectivity towards CO<sub>2</sub> in CO<sub>2</sub>/CH<sub>4</sub> mixtures, attesting its potential for biogas upgrading, with the differential approach of being a non-powder material, easily to be handled and more effective regarding diffusional barriers.

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#### CRedit authorship contribution statement

**Coelho Juliana A.:** Data curation, Investigation, Methodology, Writing – original draft. **Silva Júnior João B. A.:** Formal analysis, Investigation, Methodology. **Alves Conceição R. F.:** Data curation, Investigation, Methodology. **Oliveira Edipo S.:** Conceptualization, Formal analysis, Investigation, Methodology, Writing – original draft, Writing – review & editing. **Guedes Jadson B.:** Formal analysis, Investigation, Methodology. **Loiola Adonay:** Conceptualization, Data curation, Formal analysis, Funding acquisition, Project administration, Supervision, Writing – original draft, Writing – review & editing. **Souza-Filho José D. V.:** Conceptualization, Formal analysis, Investigation, Methodology, Writing – original draft. **Sasaki José M.:** Conceptualization, Investigation, Supervision, Writing – original draft. **Mota Francisco S. B.:** Data curation, Investigation, Methodology, Resources, Writing – original draft. **Lima Ari C. A.:** Formal analysis, Investigation, Methodology. **Bastos-Neto Moisés:** Formal analysis, Investigation, Methodology, Writing – original draft, Writing – review & editing.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Data Availability

Data will be made available on request.

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